# Surface Characterization of LaMnO<sub>3+ $\delta$ </sub> Powder Annealed in Air

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Perovskite-type LaMnO $_{3+\delta}$ , synthesized using poly(acrylic acid) (PAA), was annealed at 400 and 700°C in air for 6–54 hr. The crystallite size, the specific surface area, the La/Mn ratio, and the catalytic activity of LaMnO $_{3+\delta}$  were measured to characterize the surface. The specific surface area decreased slightly with increasing annealing time, while the La/Mn ratio of the surface, calculated from the XPS measurements, was independent of annealing time. The catalytic activity for the oxidation of CO increased with annealing. These results suggest that annealing improved the crystallinity (regularity of the ions) of the LaMnO $_{3+\delta}$  surface. © 1995 Academic Press, Inc.

## INTRODUCTION

Recently, there has been an interest in examining the oxidation of hydrocarbons on perovskite-type oxides (1-5). La $MO_3$  (M=Cr, Mn, Fe, Co, and Ni) shows a high activity for the oxidation of carbon monoxide (CO). The catalytic activity for the oxidation of CO on LaNiO<sub>3</sub> is higher than that on NiO (6). La $MO_3$  is generally synthesized at high temperature, using a standard ceramic technique. Therefore, the oxides sinter and the specific surface area of La $MO_3$  is less than 5 m²/g (2, 3). To improve the catalytic activity, it is necessary to synthesize La $MO_3$  with a large specific surface area.

Perovskite-type  $(La_{1-x}Sr_x)MnO_3$  was synthesized at low temperature using poly(acrylic acid) (PAA), and the specific surface area of  $(La_{1-x}Sr_x)MnO_3$  was ca. 18–24 m<sup>2</sup>/g (7). Recently, we noticed that the crystal structure and the oxygen content of  $(La_{1-x}Sr_x)MnO_3$  depended strongly on the PAA concentration of the gel. Therefore, we synthesized LaMnO<sub>3+δ</sub> (x = 0) at low temperature using gels with various PAA concentrations, and examined the relation between the PAA concentration, the

crystal structure, and the oxygen content (8). In case of the low PAA concentration, the oxygen content was nearly 3.00 and the crystal structure was orthorhombic. With increasing PAA concentration, the oxygen content increased to ca. 3.17 and the crystal structure changed from orthorhombic to hexagonal.

PAA burns below 300°C as shown by a thermal analysis (DTA), and the combustion heat of PAA causes a rapid increase in temperature (7). The rapid changes in temperature caused a large number of cracks in LaMnO<sub>3+ $\delta$ </sub>, resulting in a large specific surface area (9). In the present study, LaMnO<sub>3+ $\delta$ </sub> synthesized using PAA was annealed at 400 and 700°C in air for 6–54 hr to improve the crystallinity (the regularity of the ions) of the surface. X-ray photoelectron spectroscopy (XPS) measurements were performed to determine the chemical composition of the LaMnO<sub>3+ $\delta$ </sub> surface. The specific surface area and the catalytic activity for the oxidation of CO were measured. These results will provide information for characterizing the LaMnO<sub>3+ $\delta$ </sub> surface.

#### **EXPERIMENTAL**

The preparation of LaMnO<sub>3+δ</sub> with PAA has been described in detail elsewhere (8). In the present study, the PAA concentration was  $2 \times 10^{-2}$  M. The gel was fired at 400 and 700°C in air for 6–54 hr. The phases of the samples were identified by X-ray powder diffraction (XRD) using monochromatic CuKα radiation. The oxygen content in each sample was determined by an oxygen-reduction (redox) method (8). The XPS measurements were carried out for the La3d, Mn2p, O1s, and C1s levels of the samples using MgKα radiation at room temperature. The binding energy was calibrated using the C1s level from the usual contamination. The specific surface area of the samples was determined by the BET method for nitrogen adsorption. The catalytic activities for the oxidation of

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TABLE 1 Oxygen Content  $(3 + \delta)$ , Crystallite Size  $(D_{122})$ , Specific Surface Area (S) of Annealed LaMnO<sub>3+ $\delta$ </sub>, and Rate of Reaction (R) of LaMnO<sub>3+ $\delta$ </sub> for the Oxidation of CO at 270°C

Annealing cor					
Temperature (°C)	Time (hr)	3 + δ	(nm)	$S$ $(m^2/g)$	R (cm <sup>3</sup> min <sup>-1</sup> m <sup>-2</sup> )
400	6	3.13	9.3	26.9	0.070
400	30	3.20	11.1	28.0	0.109
400	54	3.21	11.1	25.4	0.122
700	6	3.13	11.8	23.5	0.213
700	30	3.20	12.7	15.7	0.295

CO were measured at 195 to 300°C using a conventional flow system. The samples (0.2 g) were preheated at 300°C under a pure oxygen stream for 3 hr. A mixed gas of CO (1.0%),  $O_2$  (4.0%), and He (balance) was fed in a flow reactor at a flow rate of 150 cm³·min<sup>-1</sup>. The products were analyzed by gas chromatography using a column (Molecular Sieve 5A) kept at 50°C during the measurements.

#### **RESULTS AND DISCUSSION**

XRD patterns of the samples annealed at 400 and 700°C in air for 6-54 hr were indexed as the hexagonal perovskite-type structure. The relation between the oxygen content  $(3 + \delta)$  of LaMnO<sub>3+ $\delta$ </sub> and the annealing condition (temperature and time) is shown in Table 1. With an increase in annealing time, the oxygen content increased to ca. 3.20. We calculated the crystallite size  $(D_{122})$  of LaMnO<sub>3+ $\delta$ </sub> from the full width at half-maximum (FWHM) of a diffraction peak (122) using the Scherrer formula (10). The relation between  $D_{122}$  and the annealing condition (temperature and time) is shown in Table 1.  $D_{122}$  increased slightly with increasing annealing time. At a certain annealing time,  $D_{122}$  of LaMnO<sub>3+ $\delta$ </sub> annealed at 700°C was larger than that annealed at 400°C. The relation between the specific surface area of LaMnO<sub>3+ $\delta$ </sub> and the annealing condition is shown in Table 1. The specific surface area of LaMnO<sub>3+8</sub> annealed at 400°C decreased slightly with increasing annealing time. As for LaMnO<sub>3+ $\delta$ </sub> annealed at 700°C, the specific surface area decreased from ca. 23.5 to ca. 15.7 m<sup>2</sup>/g. From the results of Table 1, it is considered that the crystal growth or the sintering of LaMnO<sub>3+δ</sub> occurred upon annealing, with the result that the number of cracks on the LaMnO<sub>3+ $\delta$ </sub> surface decreased.

The XPS spectra of the La3 $d_{5/2}$ , Mn2 $p_{3/2}$ , O1s, and C1s levels of LaMnO<sub>3+ $\delta$ </sub> are shown in Figs. 1, 2, 3, and 4, respectively. A satellite peak was observed to the high binding energy side of the main La3 $d_{5/2}$  peaks by ca. 4

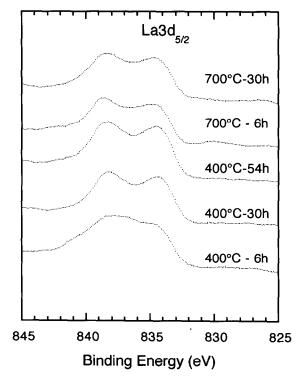


FIG. 1. The XPS spectra of the La3 $d_{5/2}$  level of LaMnO<sub>3+8</sub> annealed at 400 and 700°C in air for 6-54 hr.

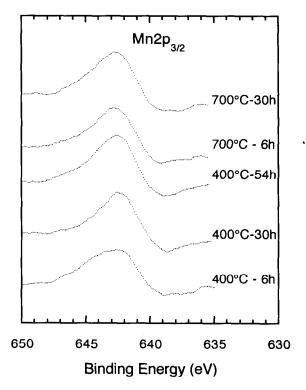


FIG. 2. The XPS spectra of the Mn2  $p_{3/2}$  level of LaMnO<sub>3+ $\delta$ </sub> annealed at 400 and 700°C in air for 6–54 hr.

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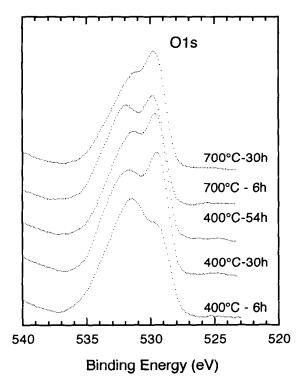


FIG. 3. The XPS spectra of the O1s level of LaMnO $_{3+\delta}$  annealed at 400 and 700°C in air for 6-54 hr.

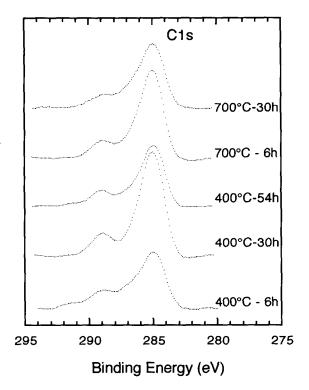


FIG. 4. The XPS spectra of the C1s level of LaMnO<sub>3+ $\delta$ </sub> annealed at 400 and 700°C in air for 6–54 hr.

TABLE 2 Binding Energies of La3 $d_{5/2}$  and Mn2 $p_{3/2}$  Levels of LaMnO<sub>3+ $\delta$ </sub>

Annealing con					
Temperature (°C)	Time (hr)	La3 <i>d</i> <sub>5/2</sub> (eV)	FWHM (eV)	Mn2p <sub>3/2</sub> (eV)	FWHM (eV)
400	6	834.9	3.1	642.5	4.6
400	30	834.3	2.2	642.5	3.6
400	54	834.5	2.5	642.4	3.8
700	6	834.7	2.5	642.8	3.9
700	30	834.7	2.9	642.8	4.0

eV. This satellite was interpreted as the excitation of an electron from the anion valence band into the lanthanum f band (11). The Mn2 $p_{3/2}$  level is broad, and asymmetric toward the high binding energy site. The binding energies of the La3 $d_{5/2}$  and Mn2 $p_{3/2}$  levels are shown in Table 2, and were independent of the annealing condition: 834.6  $\pm$  0.3 eV for the La3 $d_{5/2}$  level and 642.6  $\pm$  0.2 eV for the  $Mn2p_{3/2}$  level. Although the oxygen content of LaMnO<sub>3+ $\delta$ </sub> increased with increasing annealing time, as shown in Table 1, we could not observe a large variation in both the binding energies and FWHM of the La3 $d_{5/2}$ and Mn2 $p_{3/2}$  levels, except for that of LaMnO<sub>3+ $\delta$ </sub> annealed at 400°C for 6 hr. The O1s level has two peaks, and these peaks are independent of the annealing condition. The lower binding peak is ca. 529.7 eV and assignable to a lattice oxygen. The higher binding peak is ca. 531.5 eV and assignable to an adsorbed oxygen or hydroxyl group (12, 13). The C1s level has two peaks: one is the contaminated carbon (285.0 eV), and another small peak near 289.0 eV is considered to be carbonate (14). Because Labased perovskites are basic materials, LaMnO<sub>3+ $\delta$ </sub> is easily carbonated by moisture that yields high binding energy of O1s.

Taguchi et al. discussed the chemical bonding of orthorhombic perovskite-type (Nd<sub>1-x</sub>Ca<sub>x</sub>)MnO<sub>3</sub> with respect to the binding energy (15). The binding energies of the Nd4d, the Ca2p, and the O1s levels decreased with increasing x. The variation in the binding energy was due to the partial ionic character of Nd and Ca. In the present study, the constant binding energies of the La3 $d_{5/2}$  and the  $Mn2p_{3/2}$  levels suggest that the chemical bonding for LaMnO<sub>3+8</sub> was not affected by the annealing condition. We calculated the La and the Mn contents of the LaMnO<sub>3+ $\delta$ </sub> surface from the intensities of the La3 $d_{5/2}$  and the Mn2 $p_{3/2}$  levels (9). The relation between the La/Mn ratio of the LaMnO<sub>3+8</sub> surface and the annealing condition is shown in Fig. 5. Although the crystallite size  $(D_{122})$ and the specific surface area of LaMnO<sub>3+δ</sub> depended on annealing time, the La/Mn ratio of the LaMnO<sub>3+δ</sub> surface was independent of annealing time. With increasing an-

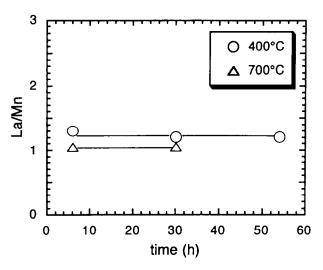


FIG. 5. The La/Mn ratio of the LaMnO $_{3+\delta}$  surface vs the annealing condition. The La/Mn ratio was calculated by the XPS measurements.

nealing temperature, the chemical composition of the LaMnO<sub>3+ $\delta$ </sub> surface approached the stoichiometric value (La/Mn = 1.0).

The conversion from CO to  $CO_2$  was measured for the various  $LaMnO_{3+\delta}$  samples. Figure 6 shows the conversion on  $LaMnO_{3+\delta}$  annealed at  $400^{\circ}C$  for 6 and 54 hr. Although the specific surface area of  $LaMnO_{3+\delta}$  decreased slightly with increasing annealing time as seen in Table 1, the catalytic activity of  $LaMnO_{3+\delta}$  annealed for 54 hr was larger than that annealed for 6 hr. These results indicate that there is an important factor, other than the large specific surface area, for improvement of the catalytic activity for the oxidation of CO. The rate of reaction (R) at a given temperature is calculated using the equation

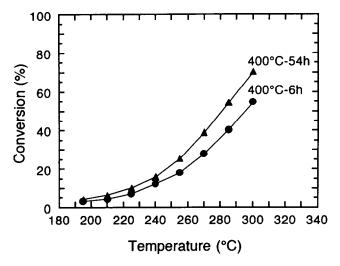


FIG. 6. The conversion from CO to  $CO_2$  on LaMnO $_{3+\delta}$  annealed at 400°C in air for 6 and 54 hr.

$$R = \frac{F \times C \times Cv}{m \times S},$$

where F is the gas flow per minute, C is the initial concentration of CO, Cv is the conversion per gram from CO to CO<sub>2</sub>, m is the mass of the sample, and S is the specific surface area of the sample (16). The relation between R at 270°C and the annealing condition is shown in Table 1. R increased with increasing annealing time or annealing temperature.

According to the results reported by Voorhoeve et al. (17), the oxidation of CO occurs at the metal ions of the surface. Both the metal ion content and the surface crystallinity play important roles in the catalytic activity. There are two kinds of oxygen at the surface: one is lattice oxygen, and the other is oxygen adsorbed on the metal ions. CO<sub>2</sub> is produced by the reaction of CO with oxygen adsorbed on the metal ions of the outermost surface. After the desorption of CO<sub>2</sub> from the surface, oxygen is again adsorbed on the metal ions. In the present study, although the Mn ion content of the LaMnO<sub>3+8</sub> surface was independent of annealing time and the specific surface area of LaMnO<sub>3+ $\delta$ </sub> decreased with annealing, the catalytic activity for the oxidation of CO increased. From these results, it is considered that annealing improved the crystallinity of the LaMnO<sub>3+ $\delta$ </sub> surface, and that the measurement of the catalytic activity is an important method for estimating the crystallinity of the LaMnO<sub>3+ $\delta$ </sub> surface. To discuss the surface crystallinity, it will be necessary to observe the LaMnO<sub>3+δ</sub> surface by high-resolution electron microscopy (HREM) in the near future.

#### CONCLUSION

The crystallite size of LaMnO $_{3+\delta}$  increased slightly with annealing. The binding energies of the La3 $d_{5/2}$ , Mn2 $p_{3/2}$ , and O1s levels from the XPS measurements were independent of the annealing condition, and the chemical bonding of LaMnO $_{3+\delta}$  was not affected by the annealing condition. The La/Mn ratio of the LaMnO $_{3+\delta}$  surface was independent of the annealing time, while the catalytic activity for the oxidation of CO increased with annealing time. These results suggest that the crystallinity of the LaMnO $_{3+\delta}$  surface is important in obtaining high catalytic activity for the oxidation of CO.

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## REFERENCES

 L. G. Tejuca, J. G. Fierro, and J. M. D. Tascon, Adv. Catal. 36, 237 (1989). 168 TAGUCHI ET AL.

- 2. T. Nitadori, S. Kurihara, and M. Misono, J. Catal. 98, 221 (1986).
- 3. K. Tabata, I. Matsumoto, and S. Kohiki, *J. Mater. Sci.* 22, 1882 (1987).
- T. Nitadori, T. Ichiki, and M. Misono, Bull. Chem. Soc. Jpn. 61, 621 (1988).
- 5. N. Mizuno, M. Tanaka, and M. Misono, J. Chem Soc. Faraday Trans. 88, 91 (1992).
- T. Seiyama, "Metal Oxides and Catalysis," p. 211, Koudan-Sha, Tokyo, 1979.
- 7. H. Taguchi, D. Matsuda, M. Nagao, K. Tanihata, and Y. Miyamoto, J. Am. Ceram. Soc. 75, 201 (1992).
- 8. H. Taguchi, H. Yoshioka, D. Matsuda, and M. Nagao, J. Solid State Chem. 104, 460 (1993).
- 9. H. Taguchi, H. Yoshioka, M. Nagao, and Y. Takeda, J. Solid State Chem. 116, 343 (1995).

- B. D. Cullity, "Elements of X-Ray Diffraction," p. 102. Addison-Wesley, London, 1978.
- 11. D. J. Lam, B. W. Veal, and D. E. Ellis, Phys. Rev. B 22, 5730 (1972).
- 12. K. Tabata, and S. Kohiki, Bull. Chem. Soc. Jpn. 65, 1295 (1992).
- Y. Wu, J. T. Mayer, E. Garfunkel, and T. E. Madey, *Langmuir* 10, 1482 (1994).
- R. Mariscal, J. Soria, M. A. Pena, and J. L. G. Fierro, J. Catal. 147, 535 (1994).
- H. Taguchi, M. Nagao, M. Shimada, Y. Takeda, and O. Yamamoto, J. Solid State Chem. 77, 336 (1988).
- Y. Ogino, T. Onoda, S. Shikagawa, M. Karube, Y. Saito, K. Tabe, T. Tamura, H. Matsumoto, M. Misono, and K. Yoshida, "Catalysis," p. 880, Maruzen, Tokyo, 1986.
- R. H. Voorhoeve, D. W. Johnson, J. P. Remeika, and P. K. Gallagher, *Science* 195, 827 (1977).